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Silver Nanowires on Carbon Nanotube Aerogel Sheets for Flexible, Transparent Electrodes

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KEYWORDS: Silver nanowires, multiwall carbon nanotubes, percolation, conductive, transparent electrode, free-standing electrode.

ABSTRACT: Flexible, free-standing transparent conducting electrodes (TCEs) with simultaneously tunable transmittances up to 98% and sheet resistances down to 11 Ω/sq were prepared by a facile spray-coating method of silver nanowires (AgNWs) onto dry-spun multiwall

carbon nanotube (MWNT) aerogels. Counterintuitively, the transmittance of the hybrid electrodes can be increased as the mass density of AgNWs within the MWNT aerogels increases, however, the final achievable transmittance depends on the initial transparency of the MWNT aerogels. Simultaneously, a strong decrease in sheet resistance is obtained when AgNWs form a percolated network along the MWNT aerogel. Additionally, anisotropic reduction in sheet resistance and polarized transmittance of AgNW/MWNT aerogels is achieved with this method. The final AgNW/MWNT hybrid TCEs transmittance and sheet resistance can be fine-tuned by spray-coating mechanisms or by choosing initial MWNT aerogel density. Thus, a wide range of AgNW/MWNT hybrid TCEs with optimized optoelectronic properties can be achieved depending of the requirements needed. Finally, the free-standing AgNW/MWNT hybrid TCEs can be laminated onto a wide range of substrates without the need of a bonding aid.

1. Introduction

The need to find materials for replacing Indium Tin Oxide (ITO) in flexible transparent conducting electrodes (TCEs) has stimulated the search for low-cost, lightweight alternatives that have high transparency and electrical conductivity, high mechanical strength, and are compatible with the numerous substrates used in optoelectronic devices. Though ITO combines high optical transmittance ($>90\%$ at 550 nm) with low sheet resistance ($10 \Omega/\text{sq}$)¹, applications of ITO-based flexible electrodes are limited by the scarcity of indium and the inherent brittleness of metal oxides^{2,3}. Novel technologies, such as for ITO nanowires⁴ and ITO mesh structures prepared via photolithography⁵, have been developed to prevent ITO from cracking. However, the fabrication of such flexible ITO TCEs requires complex and expensive processes^{1,4}. Efforts have been made to develop flexible TCEs based on multi-walled carbon nanotubes (MWNTs)^{6–8}, single-walled carbon nanotubes^{6,9–16}, graphene^{17–20}, and a combination of the above^{3,10,21,22} as alternatives to

ITO⁶. Of particular interest are free-standing MWNT aerogels sheets that can be drawn in the dry state from the side of spinnable MWNT forests²³. These highly oriented aerogels sheets are porous, transparent, flexible, and conducting. When exposed to organic solvents, like isopropyl alcohol (IPA) or methanol, they densify during solvent evaporation, reducing sheet thickness from $\sim 20\ \mu\text{m}$ to $\sim 50\ \text{nm}$. MWNT aerogels can also be laminated to a wide range of substrates by making contact in the desired section with an organic solvent and allowing surface tension effects during evaporation to densify the aerogel, increasing the contact area between MWNTs and the substrate²³. However, these self-supported aerogels are highly electronically and optically anisotropic, which can be an undesired property for optoelectronic devices.

Though the use of carbon-based nanomaterials improves the flexibility of TCEs, their low conductivity and transmittance limits optoelectronic performance²⁴. Their comparatively high sheet resistances and low transmittances make them unsuitable when choices optical transmittance above 90 % and sheet resistance below $100\ \Omega/\text{sq}$ is required^{25,26}. Thus, substantial optical and electrical improvements are necessary for carbon-based nanomaterials to become viable TCEs alternatives.

By integrating silver nanowires (AgNWs) with carbon-based nanomaterials, hybrid TCEs having low sheet resistances and high transmittances, similar to ITO, can be realized^{27,28}. The optical and electrical properties of these hybrid TCEs improve when AgNWs percolation networks are formed, increasing the number of conduction pathways connecting AgNWs either to themselves or to the carbon nanomaterial^{26,29,30}. These hybrid TCEs provide direct proportionality between sheet resistance and transmittance, as the amount of AgNWs deposited defines the maximum sheet resistance that can be obtained without losing transparency of the substrate²⁶. Previous studies have shown that AgNWs can be easily combined with carbon

nanomaterials by spray-coating, spin coating, drop-casting, or roll-coating to form hybrid TCEs^{3,17,36–44,18,19,21,31–35}. Of these techniques, spray-coating is the most suitable and scalable technique since it can be **used** over large volumes with high reproducibility, while insuring homogeneous AgNWs distributions^{30,45}.

Typically, AgNWs-carbon hybrid TCEs deploy polyethylene terephthalate (PET)^{46–49}, polyethylene naphthalate (PEN)⁶, or polydimethylsiloxane (PDMS)⁵⁰ as the flexible substrate and receive a post-treatment to further optimize the transmittance and sheet resistance. Post-treatments, including thermal annealing^{51,52}, mechanical pressing⁵³, photophysical welding^{50,54}, low energy plasma⁵⁵, and electrochemical Ostwald ripening processes⁵⁶ are quite successful, but require fabricating the hybrid electrodes separated from the optoelectronic devices to prevent damage of the layers. Free-standing TCEs that can be laminated onto any device layer have been developed for single-component-based materials, such as metallic AgNWs networks⁵⁷, graphene films⁵⁸, single-walled carbon nanotubes⁵⁹ or MWNT aerogels²³. In contrast, most of the reported hybrid carbon/metal TCEs are substrate-supported. Hence, it would be valuable to fabricate free-standing AgNWs-carbon based hybrids that meet industrial requirements for TCEs and could be laminated onto any substrate surface.

In this work, free-standing carbon/metal hybrid TCEs are produced in which the optical, electrical and mechanical properties of dry-spun MWNT aerogels are combined with the attributes of AgNWs. To avoid the trade-off between transmittance and sheet resistance, free-standing MWNT aerogels are spray-coated with a solution of AgNWs in IPA. This results in simultaneous increase in transmittance and decrease in sheet resistance as the density of AgNWs increases within the MWNT aerogel. During the process of MWNT aerogel densification by evaporating organic solvents, apertures greater than the wavelength of light ($\lambda_{\text{visible light}}$), form in

the aerogel, **increasing** overall transmittance. Also, relying on the high thermal conductivity of MWNT aerogels⁶⁰, the contact resistance of the AgNWs is reduced when a fast 1–2 minute thermal annealing process is performed. As a result, free-standing AgNW/MWNT hybrids with sheet resistances $\sim 11 \text{ } \Omega/\text{sq}$ and transmittances $\sim 98 \%$ are produced that can be laminated onto targeted substrates. Thus, the described technology produces AgNW/MWNT hybrid TCEs that exceed industrial requirements for TCEs in optoelectronic devices.

2. Experimental Section

2.1. Materials. Two types of AgNWs suspensions, AgNWs-25 and AgNWs-60 with concentrations of 5 mg/mL in isopropyl alcohol (IPA) were purchased from Seashell Technologies. The mean lengths were 30 μm and 15 μm , with mean diameter of 25 nm and 60 nm, respectively. The AgNWs solutions were diluted to a concentration of 0.25 mg/mL with IPA prior to spray coating. AgNWs-130 with diameters between 50–150 nm and lengths greater than 20 μm were synthesized by a polyol method, described by Vinogradov et al⁶¹.

2.2. Silver evaporation. A 10 nm thick layer of silver was evaporated onto MWNT aerogels with a CHA-50 electron beam evaporator at a rate of 1.2–1.7 $\text{\AA}/\text{sec}$. Prior to Ag deposition, the MWNT aerogels were vapor-densified with IPA.

2.3. Synthesis of MWNT aerogels. Free standing MWNT aerogels were drawn from the sidewall of a spinnable MWNT forest synthesized by chemical vapor deposition (CVD)²³. The substrates were prepared by depositing 3 nm of iron catalyst using an electron beam (CHA-50) onto Si wafers bearing 100 nm of silicon oxide. A mixture of acetylene (116.8 sccm) and hydrogen (1354 sccm) in a He (2400 sccm) atmosphere were reacted with the substrates at temperatures between 700–730°C for 5–10 min.

2.4. Preparation of the AgNW/MWNT aerogels. Dry-spun MWNT aerogels were placed on the top of the sheet supports having a 1x1 inch rectangular aperture in the middle and copper (Cu) tape electrodes at the edges, [see](#) Figure 1a. To measure the MWNT aerogel anisotropic sheet resistance, R_s , the Cu electrodes were positioned in a parallel (\parallel) and perpendicular (\perp) manner with respect to the horizontally oriented MWNTs electrically conductive path. In a \parallel configuration, the Cu electrodes were placed along the less resistive MWNT aerogel path, whereas in the \perp configuration, the Cu electrodes were placed along the higher aerogel resistive path, [see](#) Figure 1b. Silver paste was painted on top of the contact between the MWNT aerogels and the copper electrodes to ensure sound electrical connection. The R_s of the aerogel films were calculated using:

$$R_s = \text{Resistance} \left(\frac{\text{MWNT aerogel length}}{\text{MWNT aerogel width}} \right)$$

The transmittances of the aerogel for 550-nm-light polarized \perp and \parallel to the MWNT direction were measured using the polarization mode of a UV-Vis spectrometer (Perkin Elmer Lambda 900 UV-Vis/NIR Spectrophotometer).

Before spray coating the MWNT aerogels with the AgNWs solution, the pristine MWNT aerogels were vapor densified²³ using IPA. To accomplish this, 100 mL of IPA was placed in a 500 mL beaker, covered with aluminum foil, and heated on a hot plate at 95°C for 5 minutes to produce IPA vapors. The aerogels were placed in the beaker (facing down 2–10 cm from the liquid IPA) for 5–30 seconds, and then slowly lifted from the beaker to avoid breakage, [see](#) Figure 1c–d. A Paasche SI airbrush, which was connected to an Argon or Nitrogen gas source, was used to deposit the AgNWs onto the aerogel at a set pressure of 40 psi, [see](#) Figure 1e. The airbrush was maintained vertically during the entire spray coating process and moved from left to right to ensure uniform coverage of the MWNT aerogel. The distance between the brush nozzle

and MWNT aerogels was set at 20–30 cm. Every 20 passes, the sprayed aerogels were placed on a hot plate at 95°C for 5 minutes to complete evaporation of the IPA from the MWNT aerogels. Afterwards, resistance and transmittance measurements were made, as previously described. Thermal annealing at 160°C on a hot plate for 1–2 minutes was performed to decrease the non-effective contacts between AgNWs within the MWNT aerogels and to evaporate any organic solvent residues, see Figure 1f.

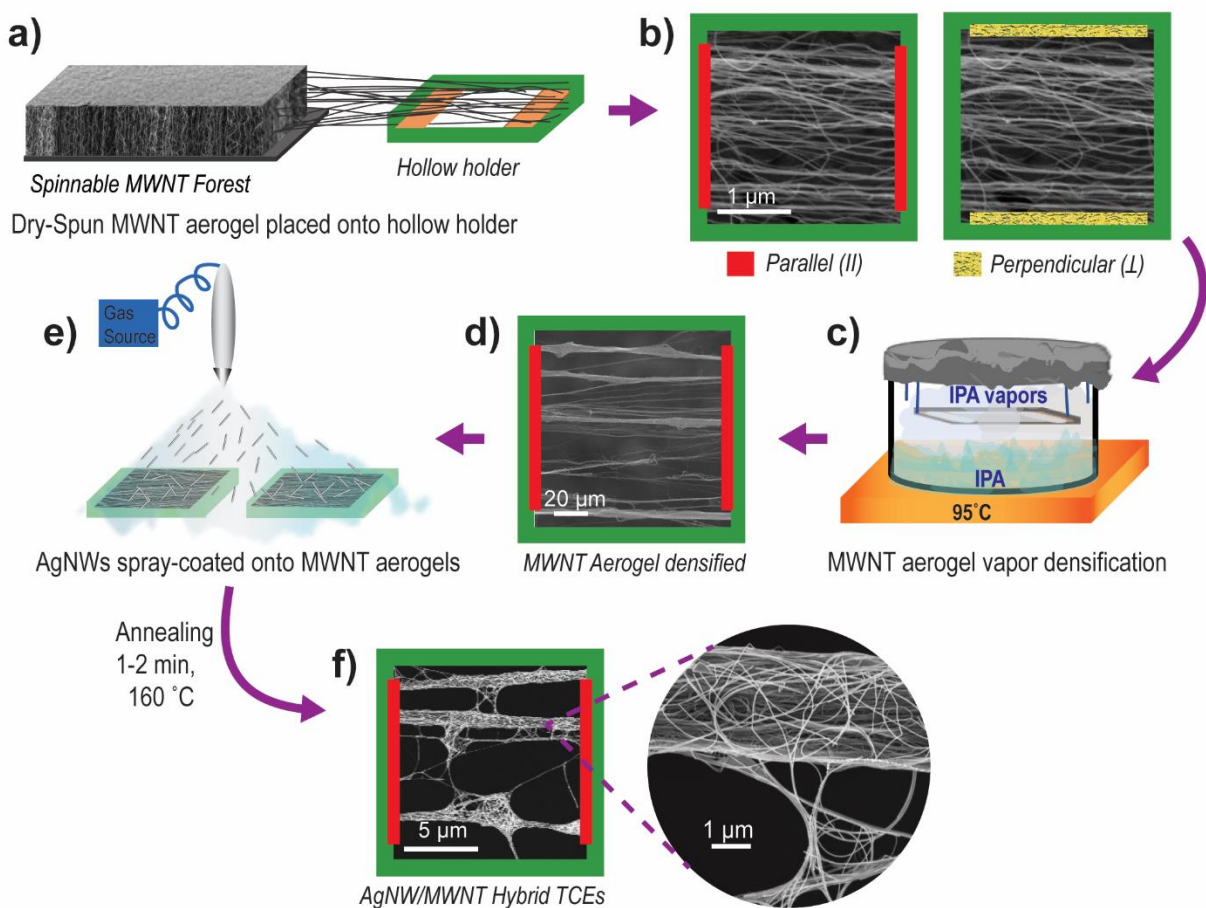


Figure 1. Preparation of flexible AgNW/MWNT hybrid TCEs. a) MWNT aerogel dry-spun on top of a hollow holder. b) MWNT aerogel parallel and perpendicular configurations. c) Densification of the MWNT aerogels with IPA vapors and d) the morphology of the MWNT

aerogel after vapor densification. e) Spray-coating AgNWs onto the perpendicular and parallel MWNT aerogel configurations and f) AgNW/MWNT hybrid TCE morphology after thermal annealing.

2.5. MWNT aerogel areal density. MWNT aerogel areal density was calculated by dividing the weight of one MWNT aerogel sheet by its area. Using measured weights of 4, 6, 8 and 10 MWNT aerogel sheets, linear regression analysis was used to predict the weight of a single aerogel. The weight of the aerogel sheets was an average of three measurements recorded with a microgram accurate scale. To minimize human error, dimensions of the sheets (width and length) were obtained using an optical microscope with attached camera and analyzed using MatLab to obtain an estimation of the area.

2.6. Physical characterization. AgNW/MWNT aerogels morphologies were characterized using a scanning electron microscope, SEM (Zeiss-LEO Model 1530 Variable Pressure Field Effect Scanning Electron Microscope) and a transmittance electron microscope, TEM (JEOL 2100F).

3. Results and discussion

Figure 2a is a photograph showing the transparency of an AgNW/MWNT hybrid TCE that was spray-coated with a 0.25 mg/mL solution of AgNWs and then thermally annealed. This hybrid TCE, which provided $T(\perp) = 98.9\%$ and $R_s(\parallel) = 11\ \Omega/\text{sq}$, is flexible (Figure S1a–b in Supporting Information), self-supporting and can be transferred to a wide range of substrates (glass, polymer, metal, etc.) without the aid of an adhesive layer. Unless otherwise indicated, the AgNW/MWNT hybrid TCEs transmittance is perpendicular transmittance, $T(\perp)$, taken at 550nm and the sheet resistance is parallel sheet resistance, $R_s(\parallel)$.

Figure 2b shows the increase in transmittance and reduction of R_s , as the amount of deposited AgNWs increases within MWNT aerogels. It highlights the three enhancing mechanisms; Densification Effect, AgNWs Effect, and Thermal Annealing Effect, which occur when spray-coating MWNTs aerogels with AgNWs/IPA solution.

The first enhancement mechanism, labeled as Densification Effect, is responsible for increasing the transmittance of the AgNW/MWNT hybrid TCEs to values $> 90\%$, and initially reducing the R_s by $\sim 30\%$. Here AgNWs account for a negligible percentage of the total decrease in R_s since a percolated AgNWs system has not been achieved²⁹. The initial decrease in R_s occurs when the solvent densifies, or reduces the aerogel initial thickness from $\sim 20\ \mu\text{m}$ to $\sim 50\ \text{nm}$, creating stronger contacts between individual nanotubes²³. SEM imaging in Figure 2c depicts the Densification Effect mechanism that occur when fine droplets of AgNWs/IPA locally wet the MWNT aerogel and IPA surface tension forces develop between parallel MWNTs strands, pushing them into thicker bundles with diameters $\leq 5\ \mu\text{m}$. As the IPA solvent evaporates, strong van der Waals forces develop between individual nanotubes within the bundles allowing them to remain collapsed and create apertures greater than $\lambda_{\text{visible light}}$, Figure 2d. While the solvent is being evaporated from the MWNT aerogel, AgNWs carried by the solvent droplets are deposited on top or between the newly formed MWNT bundles. A significant reduction of the aerogel's R_s is recorded only when the number of AgNWs reach the electrical percolation threshold. At this point, AgNWs have formed enough connecting paths with themselves, other AgNWs and/or with MWNT bundles through which electrons can travel. The effect of a simultaneous increase in transmittance and decrease in R_s does not occur when porous polymers, such as electrospun Polyacrylonitrile (PAN) nanofiber aerogel membranes, are spray-coated with AgNWs in exactly same way as for MWNT aerogels. In PAN electrospun

membranes, the R_s can be lowered (R_s final in the $k\Omega$ range) by the formation of the AgNWs conductive network around and between the apertures of the PAN fibers, in the same way that occurs in MWNTs aerogels. Because PAN fibers do not experience strong Van-der Waals bundling under surface tension forces when in contact with IPA, and its initial morphology prevents for the opening of apertures for increased light transmission, the overall transmittance decreases (contrary to the increase in MWNT aerogels) as the amount of AgNWs spray-coated to the PAN membranes increases, Figure S3a–d.

Once the AgNWs electrical percolation threshold is achieved within the MWNT aerogel, the second enhancement mechanism, labeled as AgNWs Effect in Figure 2b, takes place and the formation of the AgNWs conductive network begins. In this region, solvent densification increases the transmittance $\sim 0.1\%$ while R_s continues to decrease rapidly as more AgNWs are deposited onto the MWNT aerogel until reaching a maximum decrease in R_s of 70 %. Figure 2e portrays the final morphology of the AgNW/MWNT electrode after the AgNW network is formed. The apertures between adjacent MWNT bundles with dimensions greater than $\lambda_{\text{visible light}}$ that allow the enhanced light transmission of the AgNW/MWNT hybrid TCE are mainly formed during the Densification Effect but the AgNW conductive network, depicted in Figure 2f, is created during the AgNWs Effect. New electrical pathways are formed when AgNWs wrap around and connect MWNT bundles as well as when AgNWs connect with themselves and other AgNWs to form the AgNW conductive network lowering the R_s of the AgNW/MWNT electrode.

The R_s of the AgNW/MWNT hybrid TCE is further lowered during the third enhancement mechanism; the Thermal Annealing Effect a 1–2 minute thermal annealing process at 160°C . This quick annealing takes advantage of the high thermal conductivity of the MWNTs⁶⁰ to reduce the

non-effective contacts between AgNWs and decrease by 50 % the R_s of the AgNW/MWNT hybrids TCEs. It also evaporates any organic solvent residues from the spray-coating⁵² mechanism. Figure 2g and Figure S4a–b, show the effectiveness of this quick annealing where the AgNWs are welded to one another at the contact points without disrupting the MWNT aerogel. Overheating above the ideal temperature or for prolonged lengths of time drastically increases the R_s of the AgNW/MWNT hybrid TCEs due to fragmentation of AgNWs into shorter wires or by disintegration of AgNWs into Ag droplets, Figure S4c–d.

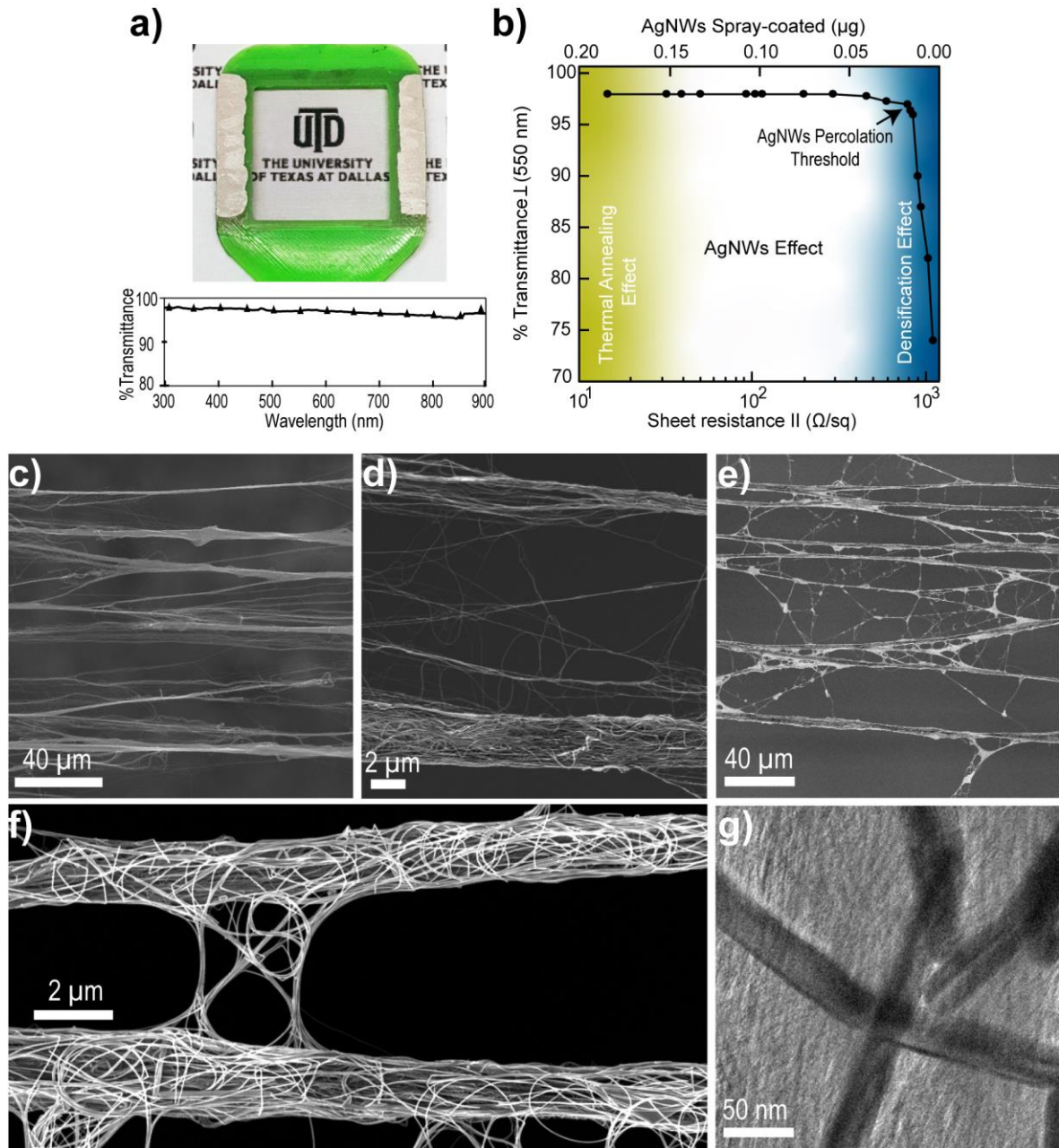


Figure 2. a) AgNW/MWNT hybrid TCE with $T_{\perp} = 98\%$ and $R_{\parallel} = 11.0 \Omega/\text{sq}$. b) AgNW/MWNT aerogel increase in T_{\perp} and R_{\parallel} reduction representing the three enhancing mechanisms occurring during spray-coating of AgNWs/IPA; Densification Effect, AgNWs Effect, and Thermal Annealing Effect. c) SEM image showing the MWNT aerogel morphology after Densification Effect and d) collapsed MWNT bundles with apertures greater than λ_{visible} light.

e) SEM image depicting the AgNW/MWNT network and f) close-up of the AgNW network formed between two MWNT bundles. g) TEM image of AgNWs welded to one another after Thermal Annealing Effect.

3.1 Optical and electrical anisotropic behavior

Dry-spun MWNT aerogels are anisotropic by nature due to the preferential alignment of nanotubes along the pulling direction. Therefore, it is expected that their optical and electrical anisotropic behavior changes when spray-coated with the AgNWs/IPA solution. A suppression of anisotropic polarized transmittance, T_A , along \parallel and \perp nanotube direction, is reduced when $T_A = \parallel/\perp$ quotient is close to unity. From Figure 3a, the AgNW/MWNT hybrid TCEs behave as a transparent isotropic material when T_A increases from 0.38 to 0.99 as a consequence of MWNT aerogel apertures greater than $\lambda_{\text{visible light}}$ formed during the Densification Effect.

The anisotropic behavior of R_s , defined as $R_{sA} = R_s(\parallel)/R_s(\perp)$, is reduced when the AgNWs electrical percolation threshold is achieved and continues to be reduced until the AgNWs conductive network is established. The R_{sA} quotient close to unity reflects a total loss of anisotropic resistance behavior and represents electrons equally percolating through the AgNW/MWNT hybrid in the \parallel and \perp MWNT directions. Before spray-coating the MWNT aerogels with the AgNWs/IPA solution, the initial aerogel $R_s(\perp)$ is ~ 70 fold higher than $R_s(\parallel)$, leading to an $R_{sA} = 0.20$. After spray-coating $0.180 \pm 0.10 \mu\text{g}$ of AgNWs, the maximum amount needed to achieve the lowest R_s , a ~ 2.5 fold difference between $R_s(\perp)$ and $R_s(\parallel)$ is recorded as $R_{sA} = 0.40$ is achieved, see Figure 3b. $R_s(\perp)$ is the limiting factor in the R_{sA} equation. For $R_s(\perp)$ to have a value similar to $R_s(\parallel)$, AgNWs need to connect adjacent MWNTs bundles and arrange

a conductive network in the perpendicular direction, requiring AgNWs with lengths \geq MWNT aerogel apertures.

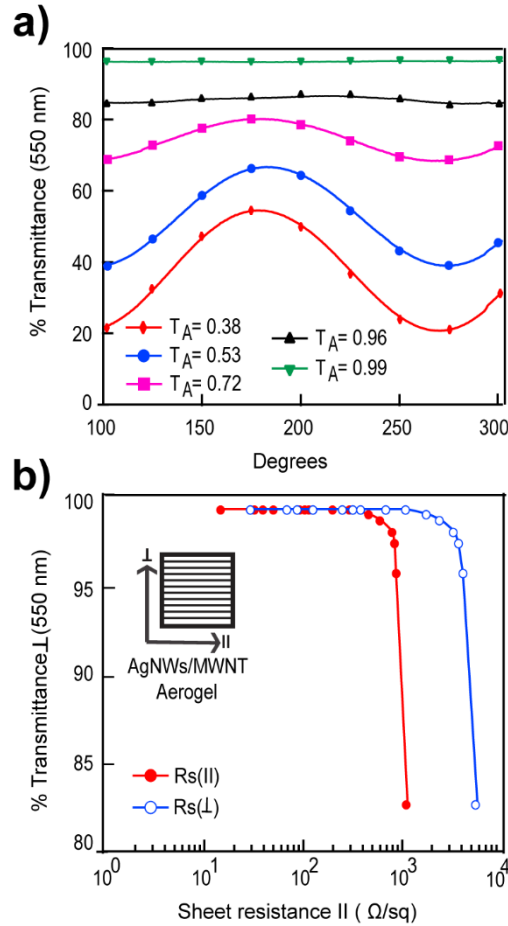


Figure 3. a) Suppression of anisotropic transmittance, T_A , taken at 550 nm while the light source rotated from 100 to 300 degrees. A maximum transmittance is obtained when \vec{E}_{field} is \perp to the MWNTs direction while a minimum is achieved when \vec{E}_{field} is \parallel to the MWNTs direction. A loss in anisotropic polarization transmittance is achieved when the transmittance anisotropic quotient, $T_A = \parallel/\perp$, increased from 0.38 to 0.99, corresponding to the sinusoidal graph profile loss. b) Plot of $T(\perp)$ at 550 nm vs R_s in the \parallel and \perp MWNTs direction. The R_s

anisotropic behavior, R_{SA} , is reduced when the quotient, $R_s(\parallel)/R_s(\perp)$, increased from 0.20 to 0.40 corresponding to a final $R_s(\parallel) = 11.0 \text{ } \Omega/\text{sq}$ and $R_s(\perp) = 29.2 \text{ } \Omega/\text{sq}$, after annealing treatment.

3.2 AgNWs dimension effects

AgNWs lengths and diameters affect the onset of the AgNWs electrical percolation threshold, the formation of the AgNWs conductive network and thus, the overall R_s of the AgNW/MWNT hybrid TCEs. Figure 4a, plots the transmittance versus R_s of three types of AgNWs with different diameters and lengths spray-coated onto MWNT aerogels. Also shown in this figure is the effect of spraying exclusively with IPA, outlining the decrease in R_s and increase in transmittance due to the Densification Effect. The highest decrease in R_s is obtained when AgNWs-25 with diameters $\sim 22 \text{ nm}$ and lengths $\sim 18 \text{ } \mu\text{m}$ are used, Figure 4a, (blue triangles). These AgNWs are flexible enough that, when sprayed, wrap around individual MWNT bundles, creating more opportunities to connect with themselves or with other AgNWs and long enough to form bridges between adjacent MWNT bundles; both are properties well suited to create the AgNWs conductive network and effectively reduce the $R_s \sim 70\%$ after densification, Figure 4c and Figure S5a–b.

On the contrary, AgNWs-60 with diameters $\geq 40 \text{ nm}$ and short lengths $< 10 \text{ } \mu\text{m}$, when sprayed to the MWNT aerogels, form non-optimal AgNWs networks and decrease the overall $R_s \sim 10\%$ after densification, Figure 4a, black squares. These AgNWs appear to be rigid and lay atop the MWNT bundles without wrapping around them, forming few bridges between adjacent MWNTs when AgNWs agglomeration occur, Figure 4d and Figure S5c–d. If the length of the AgNWs is $\geq 18 \text{ } \mu\text{m}$ and a diameter $\geq 40 \text{ nm}$ is maintained (AgNWs-130), a 50% decrease in R_s is seen after

densification, Figure 4a (red circles). In this case, AgNWs interconnect neighboring MWNT bundles, without wrapping around them, Figure 4e and Figure S5e–f.

High values of transmittances can be obtained regardless of the type of AgNWs sprayed onto the MWNT aerogels because the increase of AgNWs/MWNT TCEs transmittance is solely affected by the MWNT aerogel's solvent densification and is independent of AgNWs dimensions. The inverse proportionality between transmittance and R_s , is only observed when MWNT aerogels are sprayed-coated with an AgNWs/IPA solution, and not when silver is evaporated onto MWNT aerogels or when AgNWs/IPA are spray-coated onto glass slides. Silver evaporated onto a MWNT aerogel is a solvent-free technique and spray-coating AgNWs onto glass is unaffected by densification mechanisms thus, transmittance and R_s decrease linearly as the amount of deposited silver increases, see Figure 4b.

A summary of the lowest R_s achieved for each type of AgNWs sprayed onto MWNT aerogels, and when Ag is evaporated on top of MWNTs is presented in Table 1.

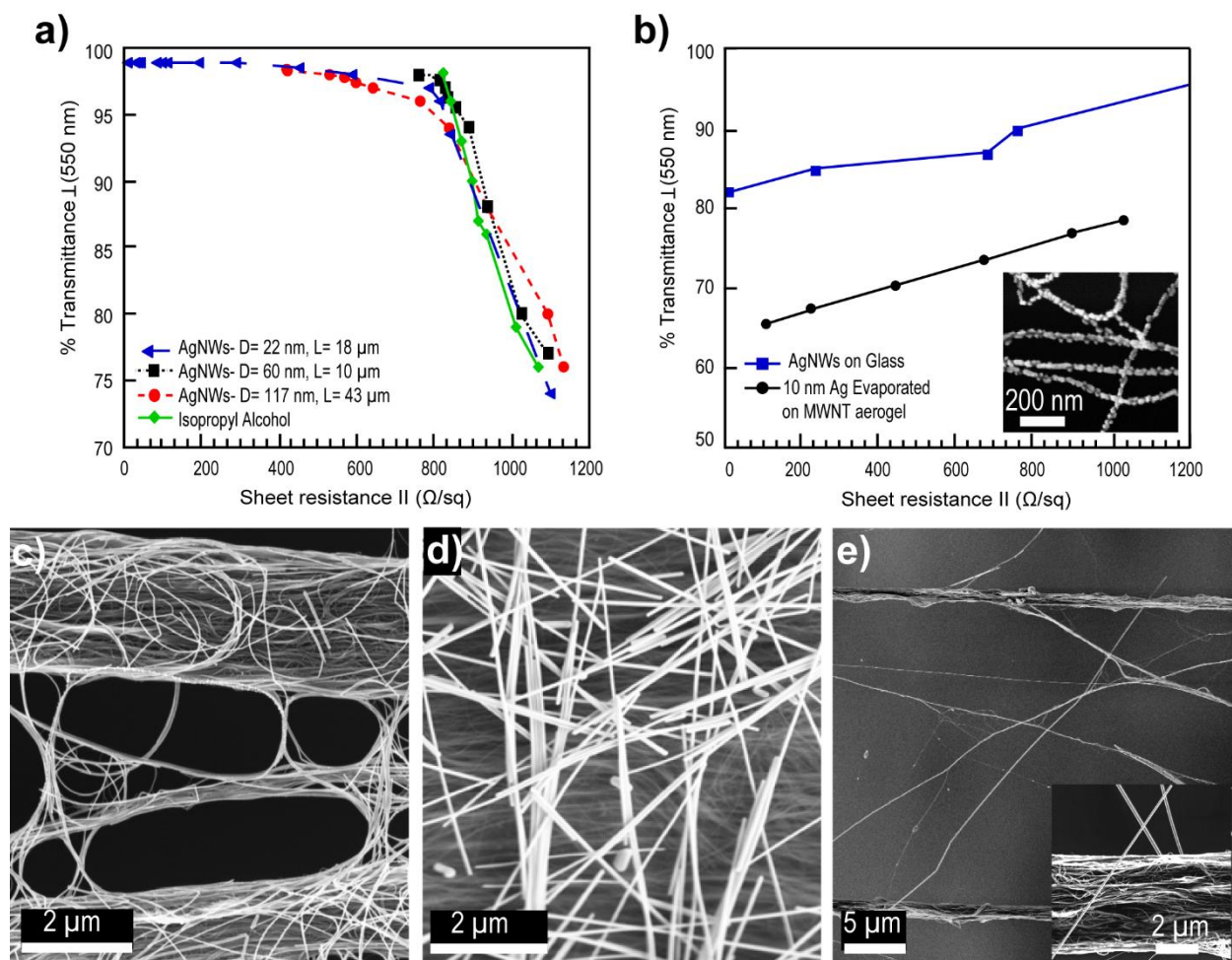


Figure 4. a) Evolution of MWNT aerogels increase in transmittance and decrease in R_s when spray-coated with AgNWs with different diameters and lengths. The effect of spraying MWNT aerogels with IPA is plotted as a reference. Linear decrease in transmittance and R_s is observed for b) 10 nm of Ag e-beam evaporated onto a MWNT aerogel, black circles and for AgNWs-25 with small diameters (22 nm) and medium lengths (18 μm), are spray-coated to a glass slide, blue squares. Morphology of MWNT aerogel with 10 nm of Ag evaporated is shown as an inset. SEM of c) AgNWs-25 with small diameters (22 nm) and medium lengths (18 μm) wrapping and connecting neighbor MWNT bundles and d) AgNWs-60 with medium diameters (60 nm) and short lengths (10 μm) laying on top of a MWNT bundle. e) AgNWs-130 with large diameters

(117 nm) and large lengths (43 μm) laying on top of a MWNT bundle without wrapping around it. All the SEM images are taken at the lowest R_s achieved before annealing.

Table 1. Effect of AgNW dimensions

AgNW/MWNT Hybrid TCEs MWNTs/()	Ave. Diameter (nm)	Ave. Length (μm)	T(\perp) max (%)	$R_s(\parallel)$ min (Ω/sq)
(AgNWs-25)	22	18	98	11
(AgNWs-60)	60	10	98	789
(AgNWs-130)	117	43	98	420
(Ag evaporated)	-	-	67	95
*AgNWs-25 on glass	22	18	82	7

3.3. MWNTs aerogel density effect

MWNT aerogel density plays an important role in constraining the maximum transmittance and minimum R_s , in the \perp and \parallel direction, that the AgNW/MWNT hybrids can reach. Ideally, low AgNWs network densities are desired to reduce the optical haze of the hybrid TCEs and still obtain low R_s and high transmittances²⁹. Although long and thin AgNWs are required to form ideal AgNWs conductive networks necessary to lower the R_s , MWNT aerogel areal density limits the amount of AgNWs per unit area needed to reach the lowest R_s and highest transmittance. Aerogels with high areal densities $\sim 5.6 \mu\text{g}/\text{cm}^2$, have initial low transmittances, $T = 53 \pm 5 \%$ and nominal sheet resistances, and $R_s = 410 \pm 25 \Omega/\text{sq}$. These aerogels require a low AgNWs mass density of $\sim 20.6 \mu\text{g}/\text{cm}^2$ to obtain a $R_s \sim 10 \Omega/\text{sq}$ after annealing, with final $T \sim 90 \%$. For MWNT aerogels with low areal densities $\sim 2.8 \mu\text{g}/\text{cm}^2$, the initial R_s and transmittances

are $825 \pm 50 \text{ } \Omega/\text{sq}$ and $70 \pm 10 \%$, respectively. With a AgNWs mass density of $\sim 50.0 \text{ } \mu\text{g}/\text{cm}^2$, these MWNT aerogels can achieve $R_s \sim 11 \text{ } \Omega/\text{sq}$ after annealing; values similar to MWNT aerogels with high areal density, but on the contrary, transmittances as high as 99 % can be achieved, see Figure 5.

A tradeoff between MWNT aerogel areal density and AgNWs mass density exists that greatly influences the optical and electrical properties of the AgNW/MWNT hybrid TCEs. Since low MWNT aerogel areal densities have less CNTs/ cm^2 , when densified with the AgNWs/IPA solution, thinner bundles and bigger apertures are formed allowing the transmittance to achieve values $> 90 \%$. The lower $< 90 \%$ transmittance, seen in MWNT aerogels with high areal densities is a consequence of more CNTs/ cm^2 available to form additional and/or thicker MWNT bundles when densified, creating apertures lower than the $\lambda_{\text{visible light}}$ that absorb or reflect the incident light more efficiently.

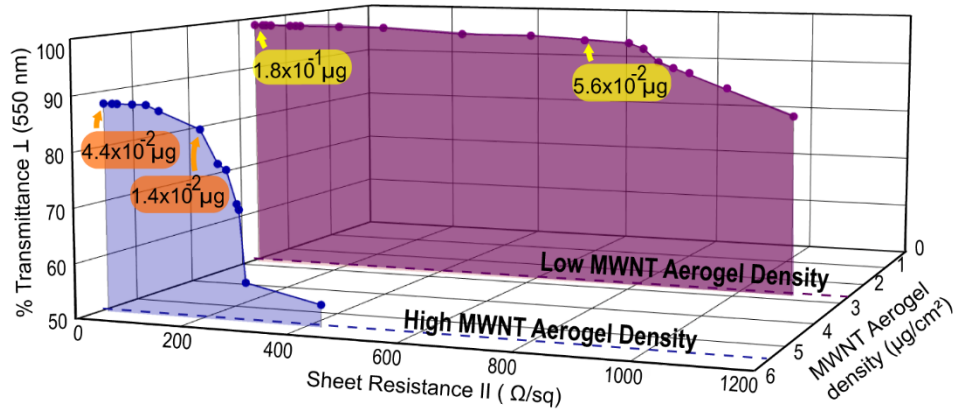


Figure 5. Transmittance \perp (550 nm) versus sheet resistance \parallel plotted as a function of MWNT aerogel areal density indicating the amount of AgNWs spray-coated (μg) After the percolation threshold was reached.

4. Conclusion

In this study, hybrid and self-supported TCEs made of AgNWs spray-coated onto MWNT aerogels have proven to achieve simultaneous increase in transmittances above 90 % and decrease in R_s below 15 Ω/sq by in situ formation of micro-sized MWNT aerogel apertures and deposition of AgNWs. AgNWs with small diameters (22 nm) and medium lengths (18 μm) were found to create ideal AgNWs conductive networks when spray-coated to MWNT aerogels by wrapping around MWNT bundles, creating effective connections with themselves and with neighboring MWNT bundles. MWNT aerogel R_s and transmittance anisotropic behavior was also suppressed by the formation of apertures with sizes larger than the wavelength of light and by creating new electrical pathways when AgNWs connected neighboring MWNT bundles. The reduced electrical anisotropy is desired for many applications in optoelectronics where charge injection or collection from the transparent electrode is expected to be uniform and isotropic in all directions.

MWNT aerogels with areal densities similar to or below 2.8 $\mu\text{g}/\text{cm}^2$, were found to be the most suitable aerogels for AgNWs deposition due to their intrinsic high transmittance. However, the high sheet resistance of these low-density aerogels required a greater amount of AgNWs ~ 50 $\mu\text{g}/\text{cm}^2$ to achieve low R_s . Of great interest is the ability to fine tune the amount of AgNWs needed within the MWNT aerogel needed to achieve the percolation critical point. This suggests further research to achieve an optimal balance of AgNWs and MWNTs in order to maximize the desirable properties of this system as a new form of TCE. In addition, these materials can easily be transferred onto various substrates, making them applicable to a wide variety of technologies within the flexible electronics field and allowing for applications not possible with traditional TCEs that are currently available.

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ABBREVIATIONS

ITO, Indium Thin Oxide; TCEs, Transparent Conducting Electrodes; MWNTs, Multiwall Carbon Nanotube; AgNWs, silver nanowires; Rs, sheet resistance; T, Transmittance; CVD, Chemical Vapor Deposition.

Associated content

Supporting information available: Bending test for the AgNWs/MWNT hybrid TCEs on PDMS substrates. TEM images of contacts between AgNWs and MWNTs and SEM images of AgNW/MWNT TCEs after ideal and beyond ideal thermal annealing conditions. SEM images depicting the morphology of AgNW/MWNT TCEs spray coated with different AgNWs-25, AgNWs-60 and AgNWs-130. SEM images of Polyacrylonitrile (PAN) membranes spray-coated with AgNWs-25.

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